Relationships among the chemical, mechanical and geometrical properties of basalt fibers

Calvin Ralph, Patrick Lemoine, John Summerscales, Edward Archer and Alistair McIlhagger

Abstract
We investigated the chemical, mechanical and geometrical properties of basalt fibers from three different commercial manufacturers and compared the results with those from an industry standard glass fiber. The chemical composition of the fibers was investigated by X-ray fluorescence spectrometry, which showed that basalt and glass fibers have a similar elemental composition, with the main difference being variations in the concentrations of primary elements. A significant correlation between the ceramic content of basalt and its tensile properties was demonstrated, with a primary dependence on the Al₂O₃ content. Single fiber tensile tests at various lengths and two-way ANOVA revealed that the tensile strength and modulus were highly dependent on fiber length, with a minor dependence on the manufacturer. The results showed that basalt has a higher tensile strength, but a comparable modulus, to E-Glass. Considerable improvements in the quality of manufacturing basalt fibers over a three-year period were demonstrated through geometrical analysis, showing a reduction in the standard deviation of the fiber diameter from 1.33 to 0.61, comparable with that of glass fibers at 0.67. Testing of single basalt fibers with diameters of 13 and 17 μm indicated that the tensile strength and modulus were independent of diameter after an improvement in the consistency of fiber diameter, in line with that of glass fibers.

Keywords
basalt fibers, chemical properties, mechanical properties, physical properties

Environmental issues, such as the waste disposal and recyclability of composites, are becoming increasingly important to both industry and governments and have led to the promotion of natural fibers as reinforcements in polymer composites. Fibers typically used as reinforcements in polymers are glass or carbon fibers due to their good mechanical properties, especially their strength; however, they are not environmentally friendly. Natural fibers, including plant fibers such as kenaf and flax, have poor mechanical properties and are prone to thermal degradation, making them uncompetitive with glass and carbon fibers. This has led to a focus on basalt fibers.

Continuous basalt fibers have a simple manufacturing process that does not require any additives. Basalt fibers are produced by melting basalt rock at temperatures between 1350 and 1700 °C and then pulling the molten material downwards through a platinum–rhodium die (bushing) using the spinneret method. The melting of basalt rock is conducted in two stages: it is first fused in the initial furnace and then transferred to the primary furnace, which controls the temperature of the melt and feeds the bushings. The fibers for processing are primarily heated by overhead gas heaters. The dark color of basalt means that material close to the surface of the melt absorbs infrared energy from the gas burners, making it difficult to obtain a homogeneous melt. There are two methods to
overcome this: holding the basalt melt in the heating stage for longer or, more commonly, using immersed electrodes to electrically heat the melt.\textsuperscript{10,11}

The chemical composition of basalt varies depending on the geographical location and conditions of formation of the source rock. Basalt consists primarily of silicon, aluminum, calcium and iron oxides, similar to glass fibers.\textsuperscript{12–14} Fibers produced from basalt consist of the minerals olivine, plagioclase, pyroxene and clino-

pyroxene.\textsuperscript{15} Basalt is classified according its SiO\textsubscript{2} content, where alkali basalts contain up to 42\% SiO\textsubscript{2}, mildly acidic basalts contain 43–46\% SiO\textsubscript{2} and acidic basalts contain >46\% SiO\textsubscript{2}. To manufacture continuous basalt fibers, the basalt rock must fall within the acidic class (>46\% SiO\textsubscript{2}).\textsuperscript{16} Recent research\textsuperscript{17} has shown that the melting properties of basalt used in manufacturing fibers varies depending on the mineral class of the basalt rock. The melting process is a crucial stage in the production of continuous basalt fibers and the homogeneity of the melt can affect the quality, diameter and performance stability of the basalt fibers. The ability to produce basalt fibers with a consistent diameter is important if these fibers are to compete with glass fibers. Significant variations in fiber diameter will affect the fiber quality, the ability to model basalt composites, the fiber volume fraction and, potentially, the interfacial adhesion through increased or reduced surface area.\textsuperscript{18,19}

Basalt fibers have superior mechanical properties to plant fibers and comparable, or better, properties to glass fibers.\textsuperscript{3,18,20} The density of basalt is between 2.6 and 2.7 g/cm\textsuperscript{3}, whereas the density of E-Glass is 2.5–2.6 g/cm\textsuperscript{3}.\textsuperscript{21} Basalt fibers have excellent sound insulation, a thermal resistance higher than that of glass, good chemical resistance to both acidic and alkaline conditions (higher than E-Glass) and are biologically inert.\textsuperscript{12,22,23} The cost of basalt fibers (about £6.00/kg) is currently higher than that of E-Glass (about £1.50/kg), although lower than that of S-Glass (about £16.00/kg). E-Glass fiber manufacturing costs have economies of scale as an established reinforcement material, whereas basalt fiber production costs are compromised by early stage small-scale production. As basalt is the most common rock on Earth, there is an abundant supply available; however, because basalt fibers require a certain SiO\textsubscript{2} content, there are currently only about three dozen mines and quarries with certified rock suitable for fiber manufacture, with the majority in Ukraine and Russia.\textsuperscript{24} The properties of basalt, together with its environmentally friendly nature,\textsuperscript{25} mean that it has potential as a competitor or replacement for glass fiber and as a new fiber in various applications. Short and continuous basalt fibers have therefore been the focus of recent research with the aim of identifying their potential applications.\textsuperscript{12,21,26–34}

With the increased demand for basalt fibers, there has been an increase in the number of established manufacturers. Glass fibers have a relatively standard performance, whereas the performance and quality of basalt fibers from different sources or manufacturers has not yet been fully examined. It is therefore important to understand the variations in basalt fibers from different manufacturers, such as the chemical composition, consistency of diameter and mechanical properties. The aim of this work was to analyze these factors and to determine whether there are any variations or relationships between them.

### Materials and methods

#### Materials

Several types of commercial basalt fiber were characterized and compared with commercially available glass fibers (Table 1). Each fiber was provided in the direct roving form with a general purpose size primarily suitable for use in epoxies. Companies A and C were chosen due to their long establishment and classification among the world leaders in basalt fiber manufacture, whereas Company B is a relatively new (5 years) and fast-emerging competitor within the market. E-Glass from Company D was selected because they are a well-established glass fiber manufacturer.

#### Methods

**X-ray fluorescence spectrometry.** The chemical composition of the fibers was determined by X-ray fluorescence (XRF) spectrometry. The fibers were initially placed in a muffle furnace at $T = 650 \degree C$ for 30 minutes to remove any sizing present on the fibers. Pyrolysis is commonly used to remove sizing and the temperatures and time used in this work were higher than those reported to be required to remove all organic sizing.\textsuperscript{35–37} After cooling, the desized fibers were milled for 2 minutes at

<table>
<thead>
<tr>
<th>Designation</th>
<th>Fiber type</th>
<th>Manufacturer</th>
<th>Nominal diameter (μm)</th>
<th>Linear density (Tex)</th>
</tr>
</thead>
<tbody>
<tr>
<td>BF1</td>
<td>Basalt</td>
<td>Basaltex</td>
<td>13</td>
<td>150</td>
</tr>
<tr>
<td>BF2</td>
<td>Basalt</td>
<td>Mafic</td>
<td>13</td>
<td>300</td>
</tr>
<tr>
<td>BF3</td>
<td>Basalt</td>
<td>GBF</td>
<td>13</td>
<td>400</td>
</tr>
<tr>
<td>BF4</td>
<td>Basalt</td>
<td>Basaltex</td>
<td>17</td>
<td>600</td>
</tr>
<tr>
<td>BF5</td>
<td>Basalt</td>
<td>Mafic</td>
<td>17</td>
<td>500</td>
</tr>
<tr>
<td>Glass fibers</td>
<td>Glass</td>
<td>PPG</td>
<td>14</td>
<td>300</td>
</tr>
</tbody>
</table>
520 rpm in a Retsch PM100 planetary ball-mill to achieve a consistent powder. The powder fiber samples were mixed with CEREOX Licowax (Fluxana, BM-0002) at a ratio of 4:1 to bind the powder and then pressed (Retsch PP25) to produce pellets for XRF analysis. CEREOX was used as a binding agent because it is clean and stable under X-rays and is designed specifically for XRF because it does not influence the results. XRF spectrometry was performed using a Thermo Scientific Niton FXL FM-XRF analyzer. Each sample was tested in three spots with a testing time of 150 s per spot.

**Analysis of fiber diameters.** Scanning electron microscopy (SEM) with a JEOL JSM-6010 microscope was used to determine the actual fiber diameter of the basalt and glass samples. Fibers were coated with gold to improve the image quality and accuracy. A set of 100 measurements was recorded from 15 mm samples taken at 1 m intervals along the roving length to give a total of 300 measurements per fiber type. Fiber sizing was not removed prior to the measurements because the calculated sizing thickness was <16 nm and therefore negligible. SEM was used instead of standard optical microscopy due to its increased image quality.

**Mechanical testing.** Single fiber tensile tests were performed according to ASTM D3379 using an Instron 5564 instrument with a 200 N load cell. As-received fibers were separated and bonded to cardboard templates, clamped in the grips of the test machine and the template was carefully cut before the start of the test. A minimum of 10 tests was performed for each sample at a constant crosshead rate of 1 mm/min for 25, 50 and 100 mm gage lengths. As it was not possible to use an extensometer or strain gage due to the small diameter of the fragile fibers, the recorded load versus displacement results were used in conjunction with the compliance method stated in ASTM D3379.

The indicated compliance was calculated using equation (1).

\[ C_s = \frac{I}{P}x(H/S) \]  

where \( I \) is the total extension for straight line section of the load–time curve extrapolated across the full chart scale, \( P \) is the full scale force, \( H \) is the crosshead speed and \( S \) is the chart speed. The true compliance is then calculated as:

\[ C = C_s - C_t \]  

where \( C_t \) is the system compliance. Young’s modulus was calculated as a corrected value using the following equation:

\[ E = \frac{L}{CA} \]  

where \( L \) is the specimen gage length and \( A \) is the average filament area.

**Results and discussion**

The chemical composition of the studied fibers is given in Table 2. The primary compound found within both the basalt and E-Glass fibers is SiO₂. The basalt fibers have a relatively consistent SiO₂ content of 48.82–49.69 mass per cent (mass%) across different manufacturers, consistent with the requirement to spin continuous basalt fibers. The glass fibers had a higher SiO₂ content of > 53 mass%, in agreement with previous studies and specifications.9,12,38,39

<table>
<thead>
<tr>
<th>Element</th>
<th>Oxide</th>
<th>BF1 (mass%)</th>
<th>Oxide</th>
<th>BF2 (mass%)</th>
<th>Oxide</th>
<th>BF3 (mass%)</th>
<th>Oxide</th>
<th>Glass fibers (mass%)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Si</td>
<td>SiO₂</td>
<td>22.52</td>
<td>48.82</td>
<td>23.22</td>
<td>49.69</td>
<td>23.26</td>
<td>49.58</td>
<td>24.78</td>
</tr>
<tr>
<td>Al</td>
<td>Al₂O₃</td>
<td>6.79</td>
<td>12.83</td>
<td>7.12</td>
<td>13.45</td>
<td>6.11</td>
<td>11.54</td>
<td>5.91</td>
</tr>
<tr>
<td>Ca</td>
<td>CaO</td>
<td>4.50</td>
<td>6.02</td>
<td>4.51</td>
<td>6.03</td>
<td>3.62</td>
<td>4.85</td>
<td>12.53</td>
</tr>
<tr>
<td>Fe</td>
<td>Fe₂O₃</td>
<td>5.18</td>
<td>7.41</td>
<td>5.25</td>
<td>7.51</td>
<td>4.87</td>
<td>6.96</td>
<td>0.17</td>
</tr>
<tr>
<td>Mg</td>
<td>MgO</td>
<td>2.45</td>
<td>4.06</td>
<td>2.03</td>
<td>3.36</td>
<td>3.08</td>
<td>5.10</td>
<td>1.82</td>
</tr>
<tr>
<td>Ti</td>
<td>TiO₂</td>
<td>0.56</td>
<td>1.18</td>
<td>0.58</td>
<td>1.21</td>
<td>0.43</td>
<td>0.90</td>
<td>0.05</td>
</tr>
<tr>
<td>K and Na</td>
<td>K₂O+Na₂O</td>
<td>1.12</td>
<td>2.44</td>
<td>1.20</td>
<td>2.50</td>
<td>1.67</td>
<td>2.13</td>
<td>0.27</td>
</tr>
</tbody>
</table>
boron, with the exception of some new boron-free glass fibers, but boron is not present in basalt. The glass fibers shared further oxides with basalt fibers, such as TiO₂, K₂O, Na₂O and Fe₂O₃, but in much lower quantities (<1 mass%). These results highlight the chemical differences between glass and basalt fibers, with the higher content of Fe₂O₃ contributing to the increased temperature resistance and darker color of basalt fibers. With the exception of a small variation in the SiO₂ and Al₂O₃ contents (about 1 mass%), samples BF1 and BF2 had a similar chemical composition. Sample BF3 had a similar SiO₂ content to BF1 and BF2, but varied consistently by 1–2 mass% for all other elements. A higher CaO content reduces the melting temperature of basalt and leads to easier homogenization of the melt, which known to aid fiber production. Samples BF1 and BF2 had a similar CaO content, whereas the CaO in sample BF3 was about 1.25 mass% lower, which could lead to an inhomogeneous melt unless adjustments are made to the furnace temperature.

Fibers from sample BF2 were chosen for further investigation to determine the consistency of fiber manufacture over time. Table 3 gives details of the fibers tested. The batches of fibers were manufactured about one year apart. The average measured diameter did not vary significantly between years, but a clear change in the standard deviation is evident, with an improvement from 1.33 to 0.61. This deviation clearly shows considerable improvements in the consistency of fiber manufacture. The fiber diameter is related to parameters such as the velocity of the molten material, the haul-off rate and the internal diameter of the bushing. It is believed that improvements in the melt homogeneity result in better control of the diameter of basalt fibers, as seen with glass fibers.

The improved results for sample BF2 were compared with the diameters of fibers from other manufacturers (Table 4). In addition to fibers tested in this work, the results were compared with previous studies on Technobasalt and D.S.E Group fibers (designated samples BF6 and BF7, respectively). The nominal diameter stated by the manufacturers of basalt fibers was 13 µm across all samples. Glass fibers were measured as 13.87 µm compared with the stated diameter of 14 µm, with a low standard deviation of 0.67. The diameter of basalt fibers needs to be consistent if they are to be competitive with glass fibers and to assist in the prediction and modeling of basalt composites. Figure 1 shows the distribution of fiber diameters each test fiber.

Fiber from one of the leading basalt manufacturers (sample BF1) was on average 1.16 µm larger than the specified diameter and had a higher standard deviation of 1.2. Although samples BF3 was close to its stated diameter, the standard deviation was more than double that of glass fibers. Sample BF6 fibers were >1 µm larger than specified, with a very high standard deviation of 2.9, suggesting poor consistency in fiber manufacture. These results highlight the current gap between glass and basalt fibers in terms of fiber manufacture and quality. However, the improved fiber of sample BF2...

<table>
<thead>
<tr>
<th>Manufacturer</th>
<th>Date of manufacture</th>
<th>Stated diameter (µm)</th>
<th>Measured diameter (µm)</th>
<th>Standard deviation</th>
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<tbody>
<tr>
<td>Mafic</td>
<td>February 2014</td>
<td>13</td>
<td>13.39</td>
<td>1.33</td>
</tr>
<tr>
<td></td>
<td>April 2015</td>
<td>13</td>
<td>13.43</td>
<td>1.10</td>
</tr>
<tr>
<td></td>
<td>August 2016</td>
<td>13</td>
<td>13.31</td>
<td>0.61</td>
</tr>
</tbody>
</table>

<table>
<thead>
<tr>
<th>Sample</th>
<th>Stated diameter (µm)</th>
<th>Average diameter (µm)</th>
<th>Standard deviation</th>
<th>Coefficient of variation (%)</th>
</tr>
</thead>
<tbody>
<tr>
<td>BF1</td>
<td>13</td>
<td>14.16</td>
<td>1.20</td>
<td>8.46</td>
</tr>
<tr>
<td>BF2</td>
<td>13</td>
<td>13.31</td>
<td>0.61</td>
<td>4.61</td>
</tr>
<tr>
<td>BF3</td>
<td>13</td>
<td>12.61</td>
<td>1.38</td>
<td>10.97</td>
</tr>
<tr>
<td>BF6 (Ref. 45)</td>
<td>13</td>
<td>14.1</td>
<td>2.9</td>
<td>4.76</td>
</tr>
<tr>
<td>BF7 (Ref. 45)</td>
<td>13</td>
<td>12.70</td>
<td>1.50</td>
<td>4.00</td>
</tr>
<tr>
<td>Glass fibers</td>
<td>14</td>
<td>13.87</td>
<td>0.67</td>
<td>4.84</td>
</tr>
</tbody>
</table>
showed significant improvements, with a diameter close to the stated value and, more importantly, a standard deviation of 0.61, lower than that of the glass samples. It is clear there have been some significant improvements in the manufacture and quality of basalt fibers in recent years. The larger diameter fibers of samples BF4 and BF5 had a high consistency in diameter with standard deviations of 0.83 and 0.69, respectively, although this is probably a result of the easier manufacturer of larger fibers.

The tensile strength and tensile modulus of all the 13 μm fibers are presented in Figure 2(a) and Figure 2(b), respectively. Initial observations of the tensile strength indicated that the fiber strength decreased as the fiber length increased for all fibers. This behavior is widely associated with an increase in the flaw population due to the longer fiber length and has been observed in both carbon and glass fibers.46,47  

There are two variables within these samples that may influence the mechanical properties: the fiber type/manufacturer and the fiber length. Two-way ANOVA was performed to determine the dependence of the tensile strength and modulus of the filaments on these two factors.48 The prerequisite of ANOVA to determine the equality of variances was determined by the Levene test.49 The test statistic $W$ was calculated by:

$$W = \frac{(N - k)}{(k - 1)} \sum_{i=1}^{k} N_i (\bar{Z}_i - \bar{Z})^2$$  

where $k$ is the number of different groups, $N$ is the total number of measurements, $Z_{ij} = |Y_{ij} - \bar{Y}_i|$ where $\bar{Y}_i$ is the mean of the $i$th group and $Y_{ij}$ is the value of the measured variable for the $j$th case of the $i$th group, $\bar{Z}$ is the mean of all $Z_{ij}$ and $\bar{Z}_i$ is the mean of the $Z_{ij}$ for the $i$th group. The resulting $P$ values for the tensile strength and tensile modulus were 0.23 and 0.49, which are significantly higher than the significance level ($\alpha = 0.05$).

Therefore the null hypothesis theory of standard variations can be accepted. ANOVA was then performed with the fiber type being Factor A and fiber length being Factor B. The calculated $P$ values from ANOVA were used as results considering a significance level of $\alpha = 0.05$. The null hypothesis of equal means is accepted when $P > \alpha$ and hence rejected when $P < \alpha$.

Table 5 gives the results for the results of the two-way ANOVA for tensile strength and tensile modulus. The reported $F$ value is the variation between the sample means/variation within the samples and is used for determining the $P$ value. For tensile strength, the very low $P$ value relating to the fiber length shows that variations in gage length are relevant at the 5% significance level, indicating a strong dependence of the strength on gage length. Low $P$ values for Factor A also indicate a dependence of fiber strength on the fiber type/manufacturer. Previous studies have confirmed the strong dependence of basalt fiber strength on gage length,15 but indicated that there was no dependence on fiber type. When the lower values of sample BF3 were removed from the ANOVA analysis, the corresponding $P$ value for fiber type increased to 0.5, which is in agreement with previous findings and highlights the poor mechanical performance of sample BF3 fibers. However, as sample BF3 is a commercially available fiber, it is important to include it in the
analysis and hence it can be suggested that there is a
dependence of tensile strength on fiber type.

A similar trend for tensile modulus can be seen from
ANOVA results. The low \( P \) values for both Factor A
and Factor B show that the elastic modulus depends on
both fiber type and fiber length. It has previously been
suggested\(^ {15} \) that the modulus did not depend on fiber
length. This change may be explained by the gage
lengths used during testing, which previously focused
on 10–40 mm. When the values for the 100 mm gage
length were removed from the ANOVA analysis, the
corresponding \( P \) value for fiber length increased to
0.16, indicating that the tensile modulus across different
fiber lengths was not significantly different. However,
comparable testing performed on E-Glass fibers\(^ {43} \) with
lengths of 5–80 mm showed that tensile modulus
increased as the fiber length increased, in agreement
with the results found for longer basalt fibers. This
increase, despite the modulus correction, can be attrib-
uted to the dependency of the test equipment on the
sample gage length. This dependency is manifested as a
contribution to elastic deformation from the testing
equipment and is in agreement with the work of
Pardini and Manhani\(^ {47} \) who reported an increase in
modulus with gage length for both glass and carbon
fibers with the ASTM correction and rigidity methods.
Comparisons between glass and basalt fibers show that
basalt is characterized by a higher tensile strength and a
comparable elastic modulus to that of glass. It is noted
that the mechanical properties are lower than the values
stated in the technical data sheet.

The tensile data was further analyzed by applying
Weibull statistics. Data for each fiber and each
gage length were sorted in ascending order. From
this, the corresponding value of the cumulative failure
probability, \( P_F \), was determined using the median rank
estimator\(^ {50} \):

\[
P_F = \frac{i - 0.3}{N + 0.4}
\]

where \( i \) is the \( i \)-th term of total number of tests \( N \).
The Weibull parameters \( m \) (shape) and \( \sigma_o \) (scale) were
determined for each fiber manufacturer and gage length
by fitting the data points with the two-parameter
Weibull distribution in equation (6):

\[
\ln[-\ln(1 - P_F)] = m \ln(\sigma) - \ln(\sigma_o)
\]

Figure 3 shows the Weibull plots obtained from
equation (6) for sample BF2 fibers and Table 6 gives
the parameters \( m \) and \( \sigma_o \) for all fibers and lengths. The
lower values of \( m \) for sample BF3 suggest that the flaws
are less evenly distributed throughout the fiber, result-
ing in a greater scatter in strength\(^ {47,50,51} \). Samples BF1
and BF2 have similar values, with the exception of
100 mm lengths, where the \( m \) value for BF1 is consid-
erably lower than that for BF, indicating a less homo-
geneous material over longer lengths.

As the Weibull parameters were obtained at different
gage lengths, it is possible to predict the tensile strength
at lengths outside the experimental range\(^ {52} \). This can be
achieved using equation (7), in particular at a cumula-
tive probability failure \( P_F = 0.5 \).

\[
\sigma = \sigma_o \left[ \frac{1}{A_o L_f} \ln^2 \right]^{1/m}
\]

where \( A_o \) is the cross-sectional area and \( L_f \) is the gage
length of the fibers. The resulting plot obtained using
the parameters from Table 6 are shown in Figure 4. The
predictions from the Weibull statistics for samples BF1

<table>
<thead>
<tr>
<th>Sample</th>
<th>Degrees of freedom</th>
<th>( F )</th>
<th>( P )</th>
<th>( F )</th>
<th>( P )</th>
</tr>
</thead>
<tbody>
<tr>
<td>Factor A (fiber type)</td>
<td>2</td>
<td>8.48</td>
<td>0.0364</td>
<td>9.46</td>
<td>0.0305</td>
</tr>
<tr>
<td>Factor B (fiber length)</td>
<td>2</td>
<td>28.28</td>
<td>0.0044</td>
<td>16.38</td>
<td>0.0118</td>
</tr>
<tr>
<td>Interaction</td>
<td>4</td>
<td>18.38</td>
<td>0.0077</td>
<td>12.92</td>
<td>0.0147</td>
</tr>
</tbody>
</table>

Figure 3. Weibull plot for sample BF2 fibers.
and BF2 are very similar, with the exception of sample BF3, in agreement with the ANOVA results in Table 5, highlighting that there may be a difference between the strength of fibers from different manufacturers.

The mechanical properties of 17 μm fibers from Company B (BF5) and A (BF4) are shown in Figure 5. The 17 μm fibers show the same trend as the 13 μm fibers in that the tensile strength increases as the fiber length decreases and the tensile modulus increases as the length is increased.

The Weibull statistics were performed again for the 17 μm fibers. The \( m \) and \( \sigma_0 \) Weibull parameters are shown in Table 7, whereas the prediction of strength at different lengths from equation (7) is presented in Figure 6.

Unlike the 13 μm fibers, there was a notable difference in strength between the 17 μm fibers in samples BF4 and BF5. Sample BF4 had a consistently lower \( m \) value at 25 and 50 mm gage lengths and was comparable at 100 mm, indicating that sample BF5 had better homogeneity. The \( m \) value decreased at 100 mm length for both fibers, confirming that critical fiber flaws are more likely to be encountered at longer gage lengths. The difference in performance between samples BF4 and BF5 is shown in Figure 6.

It has widely been thought that the tensile strength and modulus of natural fiber increases as the fiber diameter decreases. This was shown by fibers

**Table 6.** Weibull parameters for strength of 13 μm basalt fibers

<table>
<thead>
<tr>
<th>Fiber</th>
<th>( \sigma_0 ) (MPa)</th>
<th>( m )</th>
<th>( \sigma_0 ) (MPa)</th>
<th>( m )</th>
<th>( \sigma_0 ) (MPa)</th>
<th>( m )</th>
</tr>
</thead>
<tbody>
<tr>
<td>BF1</td>
<td>2065</td>
<td>38.19</td>
<td>1942</td>
<td>31.63</td>
<td>1730</td>
<td>12.71</td>
</tr>
<tr>
<td>BF2</td>
<td>2066</td>
<td>42.52</td>
<td>1971</td>
<td>26.38</td>
<td>1765</td>
<td>25.69</td>
</tr>
<tr>
<td>BF3</td>
<td>1972</td>
<td>18.56</td>
<td>1775</td>
<td>32.18</td>
<td>1477</td>
<td>15.67</td>
</tr>
</tbody>
</table>

**Table 7.** Weibull parameters for strength of 17 μm basalt fibers

<table>
<thead>
<tr>
<th>Fiber</th>
<th>( \sigma_0 ) (MPa)</th>
<th>( m )</th>
<th>( \sigma_0 ) (MPa)</th>
<th>( m )</th>
<th>( \sigma_0 ) (MPa)</th>
<th>( m )</th>
</tr>
</thead>
<tbody>
<tr>
<td>BF4</td>
<td>1962</td>
<td>13.32</td>
<td>1797</td>
<td>28.6</td>
<td>1476</td>
<td>20.2</td>
</tr>
<tr>
<td>BF5</td>
<td>2210</td>
<td>20.2</td>
<td>2001</td>
<td>47.64</td>
<td>1634</td>
<td>15.66</td>
</tr>
</tbody>
</table>

**Figure 4.** Tensile strength of 13 μm basalt fibers as a function of gage length.

**Figure 5.** Tensile properties of 17 μm basalt and glass fibers.

**Figure 6.** Tensile strength of 17 μm basalt fibers as a function of gage length.
from Company A, where there was a clear decrease in tensile strength and tensile modulus as the fiber diameter increased [Figure 7(a) and 8(a)]. By contrast, the tensile properties of glass do not depend on the fiber diameter as a result of improvements in the consistency of manufacture of glass fibers. Comparisons of tensile strength and tensile modulus between the 13 and 17 μm basalt fibers from Company B (samples BF2 and BF5) can be seen in Figures 7(b) and 8(b).

The tensile strength is nearly constant for the two fiber diameters of samples BF2 and BF5, with the exception of the longer 100 mm lengths, where a slight reduction in strength is seen at larger diameters. The cause of this difference is unknown, although there are more likely to be critical fiber flaws in longer fiber lengths, which may be more prominent at larger diameters. The tensile modulus for sample BF5 showed little deviation, suggesting that tensile strength is independent of fiber diameter.

The independence of fiber strength and diameter for samples from Company B is in agreement with previous work. Otto showed that, when fibers of different diameters are formed under controlled, near-identical conditions, their break strengths are identical and hence are reliant on the process of formation rather than the diameter. This applies to diameters >9 μm. With the demonstrated increase in the quality of basalt fibers from Company B, basalt fibers are shown to behave in a similar manner. These findings apply only to fibers on their own and not fibers embedded in a polymer matrix. Fibers tows consisting of fibers with a smaller diameter, but constant weight, have an increased surface area, which, in turn, generates more interaction and adhesion to the matrix and results in a higher mechanical performance. However, as the fibers begin with the same mechanical properties, it is thought that the effect of surface area may not be as large as for fibers that have a different performance at varying diameters.

The mechanical properties of basalt and glass fibers have been related to their chemical composition. Attempts have therefore been made to improve the mechanical properties of basalt fibers through the addition of extra elements during manufacturing, resulting in positive improvements. A relationship between the ceramic-like content (SiO₂ + Al₂O₃), which is the primary composition of basalt, and the mechanical properties has been demonstrated; however, a correlation
with the Al2O3 was not seen. Figure 9 shows the relationship between the tensile strength and the ceramic-like and Al2O3 contents.

Although glass fibers are shown on the same graph in Figure 9, they were not included in the correlation due to their different chemical composition. There is a clear correlation between tensile strength and the ceramic-like content (Figure 9(a)), but also a significant relationship with the Al2O3 content (Figure 9(b)). Two-way ANOVA was performed for tensile strength with the ceramic content as Factor A and the Al2O3 content as Factor B. The ceramic content generated a P value of 0.001, below the level of significance (α = 0.05). The resulting P value for Al2O3 was considerably lower at $1.3523 \times 10^{-7}$, suggesting that the tensile strength is more dependent on the Al2O3 content. Comparisons with the tensile modulus (Figure 10) indicate that there is no significant correlation between the modulus and the ceramic-like or Al2O3 content. Similar comparisons of mechanical properties with other elements found within basalt fibers yielded no evident relationship, suggesting they have a low importance in directly determining the mechanical properties of fibers.

**Conclusions**

The chemical composition, fiber diameter and mechanical properties of different basalt fibers were investigated using XRF spectrometry, SEM and tensile testing. The main components of the basalt fibers were SiO2, Al2O3, CaO, MgO and Fe2O3, with small amounts of TiO2, K2O and Na2O. The glass fibers had similar chemical components/constituents to basalt, with the main difference in composition being higher levels of Fe2O3 in basalt. The chemical composition of basalt remained largely consistent between manufacturers, with only sample BF3 showing a variation in Al2O3, CaO and MgO content. The diameter of the basalt fibers varied between manufacturers, with most showing a higher standard deviation than glass. Significant improvements in the distribution of fiber diameters was demonstrated for the first time, with sample BF2 being comparable with the glass fiber standard, suggesting advancements in the manufacturing quality of basalt fibers.

The mechanical properties of basalt fibers vary between manufacturers, although the properties of fibers from Company A and Company B were
comparable. The basalt fibers were characterized by a higher tensile strength than the E-Glass fibers and a similar tensile modulus. ANOVA was used to show the dependence of fiber strength on gage length, with shorter fiber lengths yielding a higher tensile strength and fiber lengths >50 mm yielding a higher tensile modulus.

For most of the commercial basalt fibers tested, the properties of the basalt fibers was dependent on the fiber diameter. Contrary to common belief, the strength and modulus of basalt fibers was independent of the fiber diameter for fibers from Company B, with fibers ranging from 13 to 17 μm diameter displaying comparable properties. A clear correlation between the mechanical properties and the chemical composition of basalt fibers was evident, with fibers showing a strong dependence on the ceramic-like content (SiO2 + Al2O3), but primarily the Al2O3 content, confirmed by ANOVA. Basalt fiber technology has reached a point where adoption should no longer constrained by product variability. The cost and performance of fibers currently lies between those for E-Glass and S2-Glass. The wider adoption of basalt fibers as reinforcement in composites will require mass production to meet the demand for fibers and should lead to their costs becoming competitive with the established E-Glass reinforcement.

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