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Analyses of the Mechanical, Electrical and Electromagnetic Shielding Properties of Thermoplastic Composites Doped with Conductive Nanofillers

*Ali Can Yilmaz^a, Mustafa Sabri Ozen^b, Erhan Sancak^b, Ramazan Erdem^c, Ozlem Erdem^d,
Navneet Soin^e

^aDepartment of Motor Vehicles and Transportation Technologies, Adana Vocational School of Higher Education, Cukurova University, 01330, Adana, Turkey

^bDepartment of Textile Engineering, Technology Faculty, Marmara University, 34365, Istanbul, Turkey

^cDepartment of Textile Technologies, Serik G-S. Sural Vocational School of Higher Education, Akdeniz University, 07058, Antalya, Turkey

^dDepartment of Textile Technologies, Adana Vocational School of Higher Education, Cukurova University, 01330, Adana, Turkey

^e Institute for Materials Research and Innovation (IMRI), University of Bolton, Bolton, UK

Abstract

The purpose of this study is to observe effect of incorporating vapor grown carbon nanofibers with various amounts in polyvinylidene fluoride matrix in terms of mechanical strength and electromagnetic shielding effectiveness. Thermoplastic conductive nanocomposites were prepared by heat pressed compression molding. Vapor grown carbon nanofibers were utilized at various weight ratios (1 wt. %, 3 wt. %, 5 wt. % and 8 wt. %) as conductive and reinforcing materials. Polyvinylidene fluoride was used as a thermoplastic polymer matrix. SEM analysis was conducted in order to characterize the morphology and structural properties of the nanocomposites and results revealed well dispersion of carbon nanofibers within the matrix for all concentrations. Mechanical characteristics were investigated according to standards. Findings proved that overall increments of 16%, 37,5% and 56% were achieved in terms of tensile strength, elasticity modulus and impact energy,

*: Corresponding author
E-mail: acyilmaz@cu.edu.tr

respectively where a total reduction of 44.8% was observed in terms of elongation for 8 wt.% vapor grown nanofiber matrix compared to that of 0 wt.%. Electromagnetic shielding effectivenesses of the nanocomposites were determined by standard protocol using coaxial transmission line measurement technique in the frequency range of 15–3000 MHz. It was observed that resistance, sheet resistance and resistivity of nanocomposites depicted substantial reduction with the increment of nanofiber content. Nevertheless, it was observed that nanofiber content, dispersion and network formation within the composites were highly influent on the electromagnetic shielding effectiveness performance of the structures.

Keywords: Carbon Nanofiber, Electromagnetic Shielding Effectiveness, Polymer Composites, Mechanical Properties

Introduction

Recently, there has been a tremendous effort spent on developing nanofiber/polymer composites for variety of industrial applications. One of the biggest reasons for these great attempts is the desire to obtain more functional and efficient materials compared to traditional microfiber/polymer composites. Inorganic fibers (glass and carbon fibers) and aromatic organic fibers (Aramid) are the traditional fillers used to increase the performance characteristics of polymers. On the other hand, it has already been reported that as the fiber diameter decreases, higher reinforcing capabilities are gained due to decrease in number of defects, increase in the contact area between filler and polymer matrix, and also increment in the fiber flexibility resulting better bending property without breaking [1]. Therefore, utilizing nanofibers as filler materials for polymer composite production has been widely investigated [2].

**: Corresponding author
E-mail: acyilmaz@cu.edu.tr*

Concerns about electromagnetic interference (EMI) have been growing among the researchers due to their potential health hazards on living creatures and their negative impacts on the performance of the electronic devices, as the usage of electrical and electronic equipment have been accumulating day by day [3]. The source of the EMI is principally electrical with unwanted electromagnetic emission being radiated or conducted [4]. Electromagnetic shielding effectiveness (EMSE) is described as blocking the electromagnetic fields by facilitating conductive or magnetic materials as barriers [5]. Effectiveness of a shield is influenced by the frequency of the EM, the distance of the shield from the source, the thickness of the shield and the shielding material composition. EMSE is calculated by (1) and evaluated in decibels (dB);

$$SE = 10 \lg \frac{P_o}{P_t} = 20 \lg \frac{E_o}{E_t} = 20 \lg \frac{H_o}{H_t} \quad (1)$$

where P_o , E_o and H_o are the power, the electric and the magnetic field intensities pertain to the shield, respectively and P_t , E_t and H_t are the factors transmitted through the shield [6]. Accordingly, some of the grades and performance specifications of the textile materials having electromagnetic shielding properties are presented in Table 1.

Table 1. Performance specifications of electromagnetic shielding textiles in general and professional use [7]

Grade	Percentage of Electromagnetic Shielding (ES)	Shielding Effectiveness (SE) in General Use	Shielding Effectiveness (SE) in Professional Use
5 Excellent	SE>99.9%	SE>30dB	SE>60dB
4 Very Good	99.9%≥SE>99%	30dB≥SE>20dB	60dB≥SE>50dB
3 Good	99%≥SE>90%	20dB≥SE>10dB	50dB≥SE>40dB
2 Moderate	90%≥SE>80%	10dB≥SE>7dB	40dB≥SE>30dB
1 Fair	80%≥SE>70%	7dB≥SE>5dB	30dB≥SE>20dB

*: Corresponding author
E-mail: acyilmaz@cu.edu.tr

Metals are stated as the most effective electromagnetic shielding materials; however, they are expensive and heavy. Furthermore, thermal expansion, oxidation and corrosion are remarkable problems during usage [8]. Polymer composites, on the contrary, are more attractive than metal coated and plated materials for variety of applications (defense, electrical and electronic etc.) due to their low weight, low cost, ease of processing and shaping, and corrosion resistance. In the last few years, conductive nanofiller/polymer composites have been widely explored due to their exceptional multifunctional properties (mechanical, electrical, thermal etc.) compared to the conventional conductive polymer composites (CPCs) which includes conductive fillers such as carbon fibers (CF), carbon black (CB), metal and ceramics. These composites have been generally prepared using high aspect ratio 1-D conductive nanofillers that include carbon nanotubes (CNTs), vapour grown carbon nanofibers (VGCNFs) and metal nanowires (MNWs) [9-11]. VGCNF increases the thermal stability of many polymers due to the high thermal stability of the nanofiber and restriction of polymer chain movement imposed by the nanofibers. High thermal conductivity and high aspect ratio of VGCNF give thermally conductive polymer composites at lower filler loading compared to conventional carbon fillers [12]. Polymers filled with conventional fillers have very limited applications in the EMI shielding market because of having less shielding capabilities.

The higher aspect ratio leads to better mechanical and shielding properties. Low electrical percolation threshold concentration can be obtained with the high aspect ratio of conductive nanofillers such as carbon based nanofibers. The high conductivity along with the high aspect ratio ensures that these nanofibers incorporated composites may have an adequate level of shielding at relatively low filler loading compared to the composites including nanoparticles or nano clays [12]. In addition, the fiber content of a composite structure has an evident influence on the composite failure strain and strength. In general, as the fiber content

**: Corresponding author
E-mail: acyilmaz@cu.edu.tr*

increases within the matrix, tensile strength also increases [13-20]. Failure mechanism researches [15,21-23] revealed that failure of the structure under tensile stress begins with the ruptures at the fiber tips due to stress concentration and propagation of these cracks along the fiber-matrix interface. As a result, it can easily be inferred that fiber concentration and naturally number of fiber tips are of great influence on determining the strength of the structure [24].

Figure 1. Schematic stacked-cup carbon nanofiber structure with a TEM image showing [54].

In this present work, VGCNFs were used as conductive and reinforcing nanofillers, and Polyvinylidene fluoride (PVDF) was utilized as polymer matrix to prepare thermoplastic composites. Although, VGCNFs were launched earlier than CNTs to the research field, much less research was carried out for VGCNFs due to smaller diameter, lower density and better mechanical properties of CNFs. It is described that VGCNFs are hollow core nanofibers composed of a single graphite layer or double graphite layers that are stacked parallel or at a certain angle from the fiber axis [24,25]. The stacked layers are nested with each other and have different formations including bamboo-like, parallel and cup-stacked [26-28]. Since VGCNFs have much lower price compared to CNFs, and especially cup-stacked form

*: Corresponding author
E-mail: acyilmaz@cu.edu.tr

contains more reactive carbon edges that can interact with the matrix better, they could be a potential alternative for manufacturing single filled, bi-filled and multi-filled synergistic reinforced composite structures for shielding purposes. The aim of this study is to prepare shielding nanocomposite structures, consist of various amounts of VGCNFs and PVDF, by compression molding, and characterize the mechanical properties as well as EMSE performance of these samples.

Experimental Work

Materials

The carbon nanofiller used in this study is graphitized VGCNF (PR-25-XT-LHT, Sigma Aldrich). PVDF was provided by Solvay. The properties of the CNFs and PVDFs are given in Table 2.

Table 2. The properties of VGCNFs and PVDF

VGCNFs		PVDF	
Type of the VGCNFs	Graphitized, platelets (conical)	Type of the PVDF	Solef ® 1000 series
Purity of CNFs	>98% carbon basis	Volume Resistivity	$>10^{14}$ ohm.cm
D (diameter)	100nm	Density	1.78 g/cm ³
L (Long)	20-200 micron	Melting Point	160-172 °C
Average pore volume	0.12 cm ³ /g	Melt Flow Index (MFI) @ 230 °C /2.16 kg	8 (g/10min)
Molecular Weight	12.01 g/mol		
Density	1.9 g/cm ³		

Preparation of Nanocomposite Structures

The VGCNFs/PVDF nanocomposites were prepared by melt compounding in twin-screw laboratory type compounder machine. The machine has six heating chambers in the extruder and an automatic feeding system. The working conditions of melt compounder are given in Table 3. At the end of the compounder extruder, the polymer strand was cooled in cold water bath and cut in pellets. Prior to compounding, the PVDF pellets and VGCNF

*: Corresponding author
E-mail: acyilmaz@cu.edu.tr

powder were placed in a vacuum oven at 80 °C and 130 °C, respectively for about 16 hours. 1 wt.%, 3 wt.%, 5 wt.% and 8 wt.% of VGCNFs were blended with PVDF and compounded together. The total amount of the masterbatches (VGCNFs/PVDF) was around 600 g. Finally, these pellets were dried in an oven for 12 hours at 80°C.

The carbon nanofiber (VGCNF) and reinforced nanocomposites were weighed to find out the amount of carbon nanofiber used within the construction. And so, for nanocomposite structures, carbon nanofiber volume fraction was calculated according to standard by using (2) [51]:

$$V_f = \frac{\rho_m \cdot W_f}{(\rho_m \cdot W_f + \rho_f \cdot W_m)} \quad (2)$$

where V_f volume fraction of fibers, W_f weight of fibers, W_m weight of matrix, ρ_f density of fibers, ρ_m density of matrix.

Table 3. Working conditions of melt compounder

Extruder Room Temperatures	164 °C; 205 °C; 210 °C; 209 °C; 209 °C; 210 °C
Feeder Speed (%)	10
Torque (%)	35
Motor Rotation Speed (RPM)	350
Delivery Pressure (Bar)	14

The VGCNF/PVDF nanocomposite plates (10x10 cm) were constructed by compression molding using Hursan compression molder (Hursan, Turkey). The compression molding conditions were as follows: temperature 195 °C, holding time 5 min and pressure 15 MPa. The properties of nanocomposites are given in Table 4.

Table 4. Properties of nanocomposite structures

*: Corresponding author
E-mail: acyilmaz@cu.edu.tr

Sample Code	Thickness (mm)	Aerial Weight (g/m ²)	VGCNF content (wt.%)	PVDF content (wt.%)	Fiber Volume Fraction
PVDF	1.97	3346.91	0	100	----
CNF/PVDF	2.14	3651.31	1	99	0.009
CNF/PVDF	2.13	3683.88	3	97	0.028
CNF/PVDF	2.14	3678.86	5	95	0.046
CNF/PVDF	2.13	3663.18	8	92	0.075

SEM Analyses of the Structures

The VGCNF/PVDF nanocomposites were characterized by Scanning Electron Microscopy (SEM, JSM-5910 LV from JEOL). First of all, samples were coated with a thin gold palladium (20/80%) layer using a sputter coater from Polaron (SC7620) and the morphology of the structures were observed by SEM analysis at an accelerating voltage of 20 kV.

Measurement of Mechanical Properties

The specimens were tested using Devotrans Universal test machine at a crosshead speed of 2 mm/min. Impact strength was checked by means of CHEAST charpy impact resistance. Charpy impact tests were performed to study the energy absorption of the nanocomposites. For each sample, five specimens were tested at room temperature and the average of the five samples was taken as the final result [29,30].

Electromagnetic Shielding Effectiveness Measurements (EMSE)

*: Corresponding author
E-mail: acyilmaz@cu.edu.tr

A coaxial transmission line method was used to test the EMSE of VGCNFs/PVDF nanocomposites. The specimens were prepared with a standard test size of various thicknesses. The outer ring of the specimen was 133 mm in diameter. Two specimens were required to be produced for the test, one for reference and another for load testing. Various researchers have described the detailed set-up and testing procedure using a plane-wave electromagnetic field in the frequency range of 15–3000 MHz. A network analyzer (Rohde Schwarz, ZVL) to generate and receive the EM signals and a shielding effectiveness test fixture (Electro-Metrics, Inc., EM-2107A) in Figure 2(a) were used to measure the EMSE, in dB, in this investigation. The shielding effectiveness was determined from (1), which is the ratio of the incident field to that which passes through the material where P_1 (Watts) is received power with the nanocomposite presence and P_2 (Watts) is received power without the nanocomposite presence. The input power used was 0 dB, corresponding to 1 mW [4,31].

Figure 2. (a) Set up of the electromagnetic shielding effectiveness testing apparatus; (b) and (c) specimens for reference and load, respectively.

This standard determined the shielding effectiveness of the nanocomposite using the insertion-loss method. The technique involved irradiating a flat, thin sample of the base material with an EM wave over the frequency range of interest, utilizing a coaxial transmission line with an interrupted inner conductor and a flanged outer conductor. A reference measurement for the empty cell was required for the shielding effectiveness

*: Corresponding author
E-mail: acyilmaz@cu.edu.tr

assessment (Figure 2(b)). The reference sample was placed between the flanges in the middle of the cell, covering only the flanges and the inner conductors. A load measurement was performed on a solid disk shape, which had a diameter the same as that of the flange (Figure 2(c)). The reference and the load measurement were performed on the same material. The shielding effectiveness was determined from (3), which is the ratio of the incident field to that which passes through the material [31].

$$\text{EMSE} = 10 \log \left(\frac{P_1}{P_2} \right) \quad (3) \quad A_b = 1 - T_r - R_e \quad (4)$$

$$R_e = \left| \frac{E_r}{E_i} \right|^2 = |S_{11}(\text{or } S_{22})|^2 \quad (5) \quad T_r = \left| \frac{E_t}{E_i} \right|^2 = |S_{21}(\text{or } S_{12})|^2 \quad (6)$$

The reflectance (R_e) and the transmittance (T_r) of the nanocomposites were also measured and the absorbance (A_b) was calculated using the equation (4) where, R_e and T_r are the square of the ratio of reflected (E_r) and transmitted (E_t) electric fields to the incident electric field (E_i), respectively, as shown in equations (5) and (6). R_e and T_r were obtained by the measurement of S-parameters, S_{11} (or S_{22}) and S_{12} (or S_{21}) for the reflection and the transmission, respectively [32,33]. For each of the samples, five measurements for EMSE tests were carried out. The shielding effectiveness measurements were conducted between the frequency ranges of 15-3000 MHz.

Results and Discussion

Structural Analysis of the VGCNFs/PVDF Nanocomposites

*: Corresponding author
E-mail: acyilmaz@cu.edu.tr

SEM images of PVDF with various CNF concentrations are depicted in Figure 3. Increased bulk density of the structure was clearly observed from the SEM analysis. Also, good level of VGCNFs dispersion occurred for all concentrations. The satisfactory dispersion of nanofibers within the matrix may be associated with the selection of proper process parameters and the compatibility between nanofillers and PVDF polymer matrix. It was also observed that while the fiber content increased in the matrix, fiber quantity per volume is also expected to increase which leads to higher the amount of the interconnected network among the fibers.

*Figure 3. SEM images of PVDF matrix with various CNF concentrations
(a) 1 wt% CNF, (b) 3 wt% CNF, (c) 5 wt% CNF, (d) 8 wt% CNF*

Figure 4. FTIR spectra of PVDF matrix with various CNF concentrations 1,3,5 and 8 wt%.

Mechanical Characteristics of the VGCNFs/PVDF Nanocomposites

As the weight ratio of VGCNFs increases in the matrix, tensile strength also increases where elongation is inversely proportional to VGCNFs wt%. The maximum tensile strength (~50 MPa) and minimum elongation (~7%) were obtained for 8 wt% of VGCNFs in the matrix when compared to sole PVDF with 44 MPa tensile strength. The mechanical properties of carbon nanofibers reinforced polymers depend on volume fraction of the fibers in the matrix [34]. At high VGCNFs compositions, when load is applied to the composite structure, VGCNFs separate from the matrix and strain energy is dissipated which entails impediment of failure of the matrix, finally increased tensile strength (Table 5).

*: Corresponding author
E-mail: acyilmaz@cu.edu.tr

Charpy test is a renowned method for evaluating impact toughnesses of structures. Nevertheless, it is a way of analyzing absorbed energy which encompasses sample width (W), length (L) and thickness (h). In this study, square shaped samples with 10x10 cm (LxW) and h of 2.5 mm were used. Considering impact energy to be totally absorbed by span size defined specimen volume, the impact energy per unit volume (e) may be found by [50] where U is the measured impact energy:

$$e = \frac{U}{WLh} \quad (7)$$

In Charpy impact tests, the load was applied normal to the laminates. The test bench consists of anvils where the sample is freely supported, a pendulum attached to a rotating arm. The pendulum with a defined mass is released from a defined height and hits the sample in the middle, drawing a circular trajectory; transfers kinetic energy and rises to a height. Difference between heights is directly proportional to the energy absorbed by the sample. When a composite structure is exposed to impact, elastic deformation is formed by a part of the energy. The rest of the energy is dissipated through the material and induces plastic deformation, such as fiber breakage, delamination, fiber–matrix debonding and matrix cracking. The energy absorbed by the specimens is an indication of the magnitude of damage. Five specimens were tested under constant load and pendulum height and averages of the 5 values were taken into account for each test. As the CNF concentration in PVDF matrix was increased, absorbed energy also increased in general. This can be attributed to the increment of bulk density with the increase in CNF amount. More fibers in the matrix means more energy dissipation through the structure and eventually increased energy absorption capacity (Table 5). Nevertheless, to an extent, as the VGCNFs ratio is augmented in the matrix, bulk

*: Corresponding author
E-mail: acyilmaz@cu.edu.tr

density also increases due to higher density of VGCNFs than that of sole PVDF in which load transfer from the resin epoxy to the fibers is formed. In this context, increased bulk density can reduce number of flaws in the structure and stress concentration possibility as well. Voids may be formed by the VGCNFs in the matrix which induce high surface area and much more absorbed fracture energy across the matrix [35]. As to enhanced fracture behavior, modulus of the structure also increases with the increment in VGCNFs ratio. With 8 wt% of VGCNFs, the modulus is >70 MPa, which is the maximum among all the cases (Table 5).

Table 5. *Tensile strength, Elongation, Charpy's impact fracture energy and E-modulus of CNF/PVDF nanocomposites*

CNF content (wt%)	Tensile Strength (MPa)	Elongation (%)	E-Modulus (MPa)	Charpy Impact (kJ/m²)
0	44.39	13.45	1211	37.77
1	47.13	12.02	1284	50.80
3	48.12	9.80	1366	52.02
5	49.21	9.36	1420	57.27
8	51.45	7.42	1666	59.17

EMSE Performances of the VGCNFs/PVDF Nanocomposites

Figure 6 shows EMSE values of VGCNFs/PVDF nanocomposites, consisting of different wt. % of carbon nanofibers, in the 15-3000 MHz frequency ranges. As the frequency increases, EMSE values of all VGCNFs/PVDF nanocomposites increase as well. There is an inverse relationship between wavelength and frequency. In this context, the frequency increases with decreasing of the wavelength [7, 36-39]. It is clearly seen that, as the amount of VGCNFs used in nanocomposites increases, EMSE values also increase in the 15-3000 MHz frequency ranges as a result of the decrease in electrical resistivity of the structures increase. Conductivity is an effective parameter on the electromagnetic shielding effectiveness [32, 40-43]. Electrical resistivity measurements were conducted via 4 point-probe system [52] as seen

*: Corresponding author
E-mail: acyilmaz@cu.edu.tr

in Figure 5 and the outcomes of electrical resistance and resistivity measurements are depicted in Table 7.

Figure 5. Four point probe schematics

Here;

$$d \text{ is used for } \frac{d_{dim}}{s} \quad (8)$$

$$a \text{ is used for } \frac{a_{dim}}{s} \quad (9)$$

$$V = I p_s \frac{1}{\pi} \left[\frac{\pi}{d} + \ln \left(1 - e^{-\frac{4\pi}{d}} \right) - \ln \left(1 - e^{-2\pi/d} \right) + \sum_{m=1}^{\infty} \alpha_m \right] \quad (10)$$

$$R_s = R_{avg} \cdot C \quad (11)$$

$$p_s = R_s \cdot t \quad (12)$$

$$V = I \cdot p_s \cdot \frac{1}{c} \quad (13)$$

$$\alpha_m = \frac{1}{m} e^{-2\pi(\alpha-2)m/d} \frac{(1-e^{-6\pi m/d})(1-e^{-2\pi m/d})}{(1+e^{-2\pi \alpha m/d})} \quad (14)$$

where; d_{dim} is the sample width (mm), a_{dim} is the sample length (mm), p_s is the sheet resistivity (Ω cm), C is the correction factor, α_m is the summation term, V is the voltage measured, I is the magnitude of source current (amps), R_s is the sheet resistance (Ω /square), t is the thickness (cm).

The sheet resistance (Ω/sq) is calculated by multiplying the average resistance with the correction factor C:

$$R_s = R_{avg} \cdot C$$

(15)

Then the resistivity ($\Omega.\text{cm}$) is calculated by [53]:

$$\rho = R_s \cdot t$$

(16)

Table 6. Electrical resistance and resistivity values

	Resistance (Ω)	Sheet Resistance (Ω/sq)	Resistivity($\Omega.\text{cm}$)
PVDF	4,53E+08	1,99E+09	3,92E+08
1-PVDF	4,05E+08	1,78E+09	3,80E+08
3-PVDF	3,69E+08	1,62E+09	3,44E+08
5-PVDF	3,45E+02	1,51E+03	3,24E+02
8-PVDF	1,25E+02	5,49E+02	1,17E+02

The nanocomposites with 8 wt.% of VGCNFs obtained the highest EMSE value (17.3 dB) in the 2100 MHz frequency. Findings revealed that the amount and the dispersion of VGCNFs had considerable impact on the electromagnetic shielding effectiveness results due to the increment in the numbers of conductive fiber network in the composites. Moreover, it was noticed that the EMSE performance of each composite increased almost cumulatively while the content of VGCNFs increased in the structures due to the high aspect ratio triggered the conductive network formation at even very low nanofiller loading [12].

Figure 7 and 8 illustrate absorbance and reflectance values of VGCNFs/PVDF nanocomposites produced with carbon nanofibers at different weight ratios (1 wt. %, 3 wt. %, 5 wt. %, 8 wt. %) in the 15-3000 MHz frequency ranges. The reflection and absorption of electromagnetic waves are related with the properties of conductive fillers. For instance, carbon fibers are defined as more absorbent materials rather than being reflective in terms of the attenuation of electromagnetic waves in especially increasing frequencies [44-47].

*: Corresponding author
E-mail: acyilmaz@cu.edu.tr

Therefore, it was seen that absorbance values of the VGCNFs/PVDF nanocomposites were higher than reflectance values at between 15-3000 MHz frequency range. Similar results were obtained by many researchers for different kinds of conductive nano filler materials [49]. Another reason for this result could be explained by the decrease in gaps between the fibers with increasing nanofiller loading, consequently enhancing the absorption loss [48].

Figure 6. Electromagnetic shielding effectiveness (EMSE) of CNF/PVDF nanocomposites.

Figure 7. Absorbance behaviour of CNF/PVDF nanocomposites between 15-3000 MHz frequency range

Figure 8. Reflectance behaviour of CNF/PVDF nanocomposites between 15-3000 MHz frequency range

Conclusions

In this research, it was observed that CNF ratio increment in nanocomposites yields a significant increase in electromagnetic shielding effectiveness and mechanical properties. Nanocomposites loaded with CNF above 5 wt% are suitable for electromagnetic shielding applications. The highest electromagnetic shielding found in this work was 17.3 dB for 8 wt. % VGCNFs/PVDF nanocomposite. It was found that the electromagnetic waves shielded by VGCNFs/PVDF nanocomposites with 8 wt. % VGCNFs were about 90% between 15-3000 MHz frequency range which is proper for general use. Nonetheless, due to higher density of

*: Corresponding author
E-mail: acyilmaz@cu.edu.tr

CNFs than that of sole PVDF, matrix with nano fibers showed higher bulk density which yielded load transfer from the resin epoxy to the fibers and finally enhanced energy absorption capacity. On the other hand, as the weight ratio of CNF increases in the matrix, tensile strength also increases where elongation is inversely proportional to CNF wt%. At high CNF compositions, when load is applied to the composite structure, CNFs separate from the matrix and strain energy is dissipated which entails impediment of failure of the matrix, finally increased tensile strength. In the context of Young modulus, 8 wt% intermingled CNF in PVDF matrix had approximately 1.2 times of sole PVDF. In other words, the mechanical properties of carbon nanofibers reinforced polymers strongly depend on volume fraction of fibers in the matrix. The low price of VGCNFs compared to competitive fillers such as CNTs will be one of the major determinants of their widespread commercial use in the polymer industry. VGCNF/PVDF composites have potential applications in many fields including: Electromagnetic wave shielding applications, ESD protection, batteries, sensors and automotive industry.

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**: Corresponding author
E-mail: acyilmaz@cu.edu.tr*

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