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An Investigation into the Stability of Graphitic C₃N₄ as a Photocatalyst for CO₂ Reduction

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Abstract: The increasing CO₂ concentration in the atmosphere exerts a significant influence on global warming and climate change. The capture and utilization of CO₂ by conversion to useful products is an area of active research. In this work, the photo-driven reduction of CO₂ was investigated using graphitic carbon nitride (g-C₃N₄) as a potential photocatalyst. The photocatalytic reduction of CO₂ was investigated with g-C₃N₄ powder immobilized on a glass support in a batch gas phase photoreactor. The experiments were carried out under UV-Vis irradiation at 70°C and an initial pressure of 2.5 bar. The only gas phase product detected during the irradiation of the g-C₃N₄ in the presence of CO₂ was CO, and the rate of production was observed to decrease over time. Oxygen

doped g-C₃N₄ was also tested for CO₂ reduction but had lower efficiency than the parent g-C₃N₄. Repeated cycles of photocatalytic CO₂ reduction showed a decline in the activity of the g-C₃N₄. In the absence of CO₂ some CO generation was also observed. Characterization of used and unused materials, using FTIR and XPS, showed an increase in the oxygen functional groups following UV-Vis irradiation or thermal treatment. While others report the use of g-C₃N₄ as a photocatalyst, this work highlights the important need for replicates and control testing to determine material stability.

1. Introduction

Global warming and climate change driven mainly by rising atmospheric CO₂ levels has led to political initiatives to reduce the amount of CO₂ released into the atmosphere. The 2020 package is a set of binding legislation to ensure the European Union meets its climate and energy goals for the year 2020.¹ The package sets three key targets: to achieve a 20% cut in greenhouse gas emissions (from 1990 levels), to generate 20% of European Union energy from renewables and a 20% improvement in energy efficiency during use. To enable progress toward the CO₂ target, the European commission has established funding for storage and capture for reuse.

Artificial photosynthesis represents one potential solution to CO₂ capture and utilization through the generation of fuels or platform chemicals by the reduction of CO₂. Considering the energy losses in converting green energy to fuels over direct use, there is a strong economic argument for the synthesis of chemical feedstocks as compared to fuels.² There are a range of different approaches to the utilization of solar energy for CO₂ reduction.^{3,4} One may use an electrocatalytic system, whereby the energy to drive the electrochemical reduction of CO₂ can be provided through external photovoltaic (PV) system. Also, one may use a photo-electrochemical cell (PEC) wherein a buried (PV) junction is used to drive electrochemical reduction of CO₂, or indeed a PEC which utilizes a semiconductor photoelectrode which acts to both absorb the electromagnetic energy and catalyze the electrochemical

oxidation or reduction reactions. In photocatalytic systems, the semiconductor material acts to harvest the electromagnetic energy, and both oxidation and reduction reactions take place on the surface of the same particles in close proximity. All approaches have advantages and disadvantages.⁵⁻⁹ In this work we are concerned with photocatalytic reduction of CO₂ using graphitic carbon nitride (g-C₃N₄). g-C₃N₄ consists of earth-abundant C and N elements and due to its facile synthesis, non-toxicity, 2D morphology,¹⁰ appealing electronic band structure and claimed stability,¹¹ g-C₃N₄ is considered to be a potential material for the solar photocatalytic reduction of CO₂.¹² As reported in literature, in theory the conduction band edge of g-C₃N₄ phase is sufficiently negative to reduce CO₂,^{11, 13, 14} and the valence band edge is positive enough to split water. g-C₃N₄ has a valence band (VB) potential of +1.4 and a CB potential of -1.3 eV (vs NHE).^{14, 15} g-C₃N₄ materials, are naturally rich in N basic sites, which favor weak CO₂ adsorption, compared to oxide surfaces. The photocatalytic reduction of CO₂ on C₃N₄ under UV-Vis irradiation has been reported by a number of researchers utilizing heterogeneous batch reactors. When g-C₃N₄ was used as a photocatalyst for CO₂ reduction, MeOH, EtOH and HCOOH were reported as the main products in the liquid phase, whilst CH₄ and CO were reported to be found in the gas phase.¹⁶⁻²⁰ In both liquid and gas phase photocatalytic systems, the reaction rates are very low and are reported in $\mu\text{mol g}^{-1} \text{h}^{-1}$. A moderate increase of CO₂ photoconversion for metal g-C₃N₄ composites has been reported. In order to perform the CO₂ conversion protons for these reactions are obtained from water splitting reactions, to reduce back reaction reduction and oxidation sites are often physically separated by using surface co-catalysts²¹⁻²³. The co-catalysts studied have been noble metals²⁴⁻²⁷ (Pd, Pt, Co, Ru), semiconductors²⁸⁻³⁰ (TiO₂, WO₃, CeO₂) and non-metal dopants (O, P, S).³¹⁻³³ In this work we investigate g-C₃N₄ and oxygen doped g-C₃N₄ as potential photocatalysts for CO₂ reduction.

Polymeric carbon nitride, g-C₃N₄, exists in seven different polymorphic phases, and all of them are semiconductors.^{21,22} Nevertheless only g-h-heptazine and g-h-triazine phases have suitable band gaps and band edge potentials to drive photocatalytic processes. By considering the position of the VB, only the g-h-heptazine meets the potential requirements for water oxidation.^{34, 35} As this is the only

phase with suitable band edges to drive water oxidation, this phase of g-C₃N₄ material is the most frequently investigated as a photocatalyst. In order to increase the CO₂ photoreduction activity of g-C₃N₄, many research groups have looked at doping with non-metal species to change the band structure or adsorption properties. The most frequent strategy is to dope with N by using hydrazine pre-treated, melamine precursor. Other dopants can be S and O deriving from thiourea and urea, respectively. Oxygen can be also provided by oxidizing the material with H₂O₂.³⁶ The doping has been reported to generate a second semiconductor and/or a heterojunction, which is believed to decrease the recombination of the electron-hole pairs, thereby increasing the activity.^{32, 36} Non-metal dopants result in an increase of the VB width, in combination with an elevated CB minimum and a slightly reduced absorbance. Oxygen doping in the g-C₃N₄ lattice was reported to induce intrinsic electronic and band structure modulation, leading to improved surface area, extended visible light response, and enhanced separation efficiency of the photo-generated charge carriers.³⁶ Oxygen doped materials have been reported as promising photocatalysts for CO₂ reduction.³²

Although significant works on pure and modified C₃N₄ as an efficient potential photocatalyst for CO₂ reduction have been reported, replicate testing for material stability and controls in the absence of CO₂ are often not reported,^{31, 37, 38} while some report that the material maintains a percentage of stability after a number of runs.³⁹ When the g-C₃N₄ stability is tested, the methodology relies on observing changes in product yield with time, wherein the activity is reported to be constant with reuse.^{29, 32, 40} In a recent work, Xiao et al.⁴¹ which studied the g-C₃N₄ stability in the presence of water pollutants, found an accumulation of carbon species (by total organic carbon analysis) and NO₃⁻ ions in water under photocatalytic reactions (sunlight/O₂/nanostructured C₃N₄ conditions), suggesting chemical instability of g-C₃N₄ in the aqueous phase. In this study, we describe changes in activity and photocatalyst chemistry during the photocatalytic reduction of CO₂ in the gas phase with water as the reductant and with g-C₃N₄ and an oxygen doped analogue catalyst as a function of time and reuse cycle. In addition, the source of observed CO produced during photoreduction testing was performed using control runs (studying the potential photocatalysts activity in CO₂ absence). In order to correlate

changes in photocatalytic activity with reactor history, the g-C₃N₄ chemical structure was investigated by FTIR and X-ray photoelectron spectroscopy (XPS) analyses both before and after its use in the photocatalytic reactor.

2. Experimental section

2.1 Synthesis of g-C₃N₄ and oxygen chemically modified g-C₃N₄ powder.

g-C₃N₄ was prepared by a two-stage thermal polymerization process.⁴² Firstly, melamine (10 g) was directly heated at 520°C, in a crucible with a cover, for 2 h in static air with a temperature ramp rate of 2°C/min. The resultant solid was ground with a pestle and mortar, then spread on the bottom of a ceramic bowl with a diameter of 14 cm, and then heat treated in air at 500 °C for 4 h with a temperature ramp rate of 3°C/min. The resulting yellow powder was designated g-C₃N₄.

Oxygen doped g-C₃N₄, was prepared by treat the graphitic material with H₂O₂.⁴³ Briefly, 1 g of g-C₃N₄ (as synthesized previously) was dispersed in 80 mL of 30% H₂O₂ solution and placed in a Teflon vessel of 100 mL. The vessel was then heated up to 130 °C and kept at this temperature for 24 h in an autoclave. After cooling, the dispersion was filtered, and the powder was washed with hot water to remove the H₂O₂. Once no H₂O₂ was detected in the washings (by KMnO₄ method), the powder was dried at 60°C and designated O_R-C₃N₄. Oxygen doped g-C₃N₄ was also produced by thermal treatment of pure g-C₃N₄ thin films designated O_T-C₃N₄ as described below in photocatalyst immobilization.

2.2 g-C₃N₄ and oxygen thermally modified g-C₃N₄ immobilization.

To test the materials (g-C₃N₄ and O_R-C₃N₄) for CO₂ photoreduction activity, 8 mg of powder was immobilized on borosilicate glass (3 x 2.5 cm²) by spray coating using a water dispersion of 25 g L⁻¹

of C_3N_4 . Following spray coating, the samples were dried at 70 °C. The g- C_3N_4 sample was used without further treatment or annealed at 450 °C (2 h with ramp up and down rate 2°C min^{-1}) to produce $\text{O}_\text{T}\text{-C}_3\text{N}_4$ prior to testing for CO_2 reduction activity.

2.3 Photocatalytic reduction of CO_2

Photocatalytic tests were performed in a stainless steel reactor, with an internal volume of 120 mL. The reactor was fitted with a quartz window to allow irradiation of the catalyst. The reactor set up is shown in Figure 1 and is similar to that reported by Cortes et al.⁴⁴. The catalyst samples were placed inside the reactor facing the quartz window. High purity water (1 mL) was added to the bottom of the reactor at the start of each experiment. The reactor was purged for 10 min with 20% CO_2 (BOC, UN1013, 99.99% purity) in Ar (BOC, UN1006, 99.998% purity) before sealing at a pressure of 2 bar. For the control tests in the absence of CO_2 , the reactor was instead purged for 10 min with Ar, and 1 mL of heated water was added to the reactor. The temperature of the reactor was maintained with thermal tape at 70°C for all tests. The light source was a 100 W Xe lamp (LOT Oriel) with an IR filter (light intensity reported in the support information section, see Figure S1).

Figure 1. Batch photoreactor scheme equipped with a mass flow controller and light source. The system is on line connected with a GC-FID/TCD.⁴⁴

The gaseous products obtained following periodic irradiation were analyzed by gas chromatography (GC) with a flame ionization detector (FID) and using Helium (BOC, UN1046, 99.999% purity) as a carrier gas. The GC (Agilent Technologies 7890B) was connected directly to the reactor and samples were analyzed every 30 min leading to a decrease in pressure inside the reactor (0.20 bar) each time a measurement was performed. Tests were replicated to determine the reusability of the material and were named RnCO₂ (where R means the replicate runs and n is the number cycles of CO₂ reduction testing performed). The control test in the absence of CO₂ was carried out with the same sampling time points as the other tests and was designated as RnAr. To investigate the effect of surface contaminants some of the samples were exposed to photocatalytic cleaning in the presence of oxygen prior to testing under standard CO₂ reduction conditions.

2.4 Materials characterization

A Perkin Elmer Varian 640-IR FTIR was used for FTIR characterization of the samples before and after use. Analysis was performed with a resolution of 10 cm⁻¹ step size, compiled from 50 scans and measured in triplicate. The crystalline phase structures were analyzed at room temperature by powder X-ray diffraction by using a Panalytical Empyrean, equipped with Cu K_α radiation (1.5406 Å) and PixCel-1D (tm) detector. Diffuse reflectance was used to determine the optical characteristics of the materials and measured with a Shimadzu UV-2401 PC spectrophotometer, with BaSO₄ as the reference material. A Kratos Axis Ultra system using a monochromatic Al K_α X-ray source (hν 1486 eV) was used to measure X-ray photoelectron spectra (XPS) from used and unused material plates. Triplicate analysis was performed, sampling from different areas of the material plates. For all the elements, the coefficient of variance was less than 10%. The binding energy at 284.8 eV was used for calibration of C 1s chemical shifts. Survey scans covering the entire energy range were performed with 160 eV energy window, whilst high resolution scans of specific elemental regions were

performed at a reduced 20 eV. The data were fitted using a mixed Gaussian-Lorentzian (GL30) function on a Shirley background.

3. Results and Discussion

3.1 Characterization materials

Analysis of the g-C₃N₄ and O_R-C₃N₄ powder was undertaken using, FTIR, XRD, and UV-Vis reflectance. The FTIR analysis correlates to previous reports on g-C₃N₄.^{32, 45-52} The finger printing from 807 to 1650 cm⁻¹ and from 3100 to 3500 cm⁻¹ by FTIR analysis demonstrates that the materials synthesized are C₃N₄ structure type (Figure 2). The peaks at 807 cm⁻¹ are the characteristic breathing mode of the s-triazine ring system. The peaks at c.a. 1242, 1322, 1412, 1563, and 1634 cm⁻¹ are typical stretching vibration modes of C=N and C-N heterocycles. In addition, the broad peak between 3600 and 3000 cm⁻¹ is caused by the NH and OH stretches. However C-O and C=O bonds are difficult to distinguish in O doped g-C₃N₄ by FT-IR, as the C-N, C-C and C-O bonds have very similar force constants.⁵³ Due to this constraint and the limited amount of O expected to be incorporated in the O_R-C₃N₄, no appreciable changes were observed when comparing it with the g-C₃N₄ catalyst by FTIR.

Figure 2. FTIR spectra of as synthesized g-C₃N₄ (red line) and O_R-C₃N₄ (black line) showing characteristic g-C₃N₄ fingerprint.

XRD analysis of the synthesized material showed a main 2θ peak at 27.4° which has been assigned to the (002) plane of g-C₃N₄.^{14, 54} This corresponds to a d-spacing of 0.326 nm. This peak is usually assigned to the distance between the layers of the 2D graphitic material, which undergo a little left-shift in the O doped materials, as reported in literature.⁴³ The secondary intensive peak at 2θ of 13.0° is attributed to the (100) plane, with a d-spacing of 0.680 nm, which is usually reported as the intralayer d-spacing.^{14, 54} From identification of these features in the synthesized g-C₃N₄ material and comparison to tri-s-triazine spectra from literature, the synthesized materials can be confirmed to be the h-g heptazine phase of C₃N₄. The XRD spectra of the synthesized materials compared to expected peak positions previously reported by Fina et al⁵⁴ are shown in Figure 3.

Figure 3 XRD patterns of pure g-C₃N₄ and O_R-C₃N₄ as prepared and the peak positions for two phases of g-C₃N₄ reported by Fina et al.^{35, 54} The name reported heptazine (tri-s-triazine) represents the melem unit.

To discriminate between the different phases of g-C₃N₄ and to estimate the band gap energy of the material, the optical band gap was measured. When a semiconductor absorbs light with suitable wavelength, electrons are excited from VB to CB with the simultaneous formation of holes in the VB. These charges produced by photoexcitation can reduce and oxidize the adsorbed surface species, respectively, or give rise to recombination phenomena. A recent study reports a computational method which allows to correlate the thickness of s-triazine and h-g heptazine C₃N₄ with the band gap energy of the materials.⁵⁵ Diffuse reflectance spectroscopy was utilized to determine the band gap of both synthesized g-C₃N₄ catalysts (Figure 4). For pure g-C₃N₄ the optical band gap was 2.8 eV and for the O_R-C₃N₄ the optical band gap was determined to be 2.7 eV. Both values appear consistent with a hexagonally structured h-g-heptazine C₃N₄.³⁵ The samples doped with oxygen showed a red shift of the optical band gap with respect to the pristine materials.^{14, 32, 46}

Figure 4. UV-Vis diffuse reflectance spectra of powder of pure g-C₃N₄ (red line) and O_R-C₃N₄ (black line) as prepared samples. The inset shows the C₃N₄ Kubelka-Munk function vs photon energy, giving the graphic interpolation a band gap energy value equal to 2.8.

3.2 Photocatalytic reduction of CO₂

Measurement of photocatalytic activity for CO₂ reduction was performed in a gas phase batch photoreactor testing the g-C₃N₄, O_R-C₃N₄ and O_T-C₃N₄ immobilized samples. Chromatographic determination of products from photocatalytic testing yielded only CO, within the limits of detection. No H₂, O₂ (from water splitting reactions), methane or other hydrocarbons were observed. A typical graph of CO production vs time using immobilized g-C₃N₄ is shown in Figure 5.

Figure 5. Concentration of CO measured at different time points over 5 h irradiation in the presence of CO₂ using g-C₃N₄.

The samples were retested to investigate their stability. The subsequent reuse of samples showed a decline in the CO concentration (Figure 6) with replicates (R1CO₂ and R2CO₂).

Control runs in the absence of CO₂ were undertaken. In the absence of CO₂, CO was still observed as a product, and a similar trend of lower CO generation on reuse was detected (Figure 6 - R1-R3Ar runs). The observation of only CO as gaseous product, the declining CO generation rate (on reuse or extended use), and the observation of CO in the absence of CO₂ suggest that some CO₂ molecules from atmosphere remain strongly adsorbed on catalyst surface. To confirm this behavior a batch test was carried out in inert atmosphere in absence of CO₂ (N₂ humid) at atmospheric pressure and under visible light. This test confirmed a release of CO₂ with the irradiation time as reported in Figure S2. Consequently, the adsorbed CO₂ could be responsible for the observed CO. In fact, in presence of lower CO₂ amounts (just the adsorbed one), a slower catalyst inactivation occurred (see Figure 6) with respect to the test in presence of higher CO₂ concentration (20% in Ar flow). The CO yield was slightly higher for repeat experiments undertaken in the absence of CO₂ (Figure 6). DFT studies by Wu et al suggest physisorption of CO₂ and CO leads to a buckling of the carbon nitride requiring energy input for desorption and a straightening of the underlying carbon nitride.⁵⁶ The desorption of CO in CO₂ compared to an argon atmosphere (absence of CO₂) would be limited by interactions with surface adsorbates maintaining a crumpled structure that reduces the desorption.

Figure 6. CO concentration obtained after 5 h of irradiation time in the presence (RnCO₂) and in the absence (RnAr) of CO₂ with g-C₃N₄.

To investigate if adsorbed CO₂ or organic fouling was an issue, a twice used (R2CO₂) sample was photocatalytically cleaned for 3 h in O₂ flow under UV-Vis irradiation. The photocatalytic cleaning with O₂ did not alter the material reactivity within the errors of the analytical system as shown in Figure 7.

Figure 7. CO photo-generated in the presence of CO₂, before (R2CO₂) and after cleaning (CO₂ after cleaning) obtained during 3 h of irradiation with g-C₃N₄ as potential photocatalyst.

An oxygen thermally doped carbon nitride, O_T-C₃N₄, was also tested for CO₂ reduction using the same methodology as the one utilized in g-C₃N₄ testing. During this experiment, only CO was detected, the CO generated was observed to decrease with reuse. This is a similar trend to that observed in the case of pure g-C₃N₄. Comparison of activity observed with pure g-C₃N₄ and O_T-C₃N₄ showed oxygen modification to be detrimental to CO formation (Figure 8). Carbon nitride chemically oxygen doped material, O_R-C₃N₄, was also tested under identical experimental conditions and the yield of CO detected was smaller than those found for the thermal oxygen doped and unmodified C₃N₄ samples.

Figure 8. Reproducibly test of CO production in the presence of g-C₃N₄ and O_T-C₃N₄ after 5 h of irradiation time in the presence of CO₂.

Table S1 reports the amount of photogenerated CO expressed in terms of both, ppm and $\mu\text{mol g}^{-1}$ catalyst. The apparent Quantum Yield, calculated according to Tan et al.,⁵⁷ is also reported. Notably, the yields decrease with reuse for all catalysts tested.

Given the limited range of products detected under irradiation, the reduction in product yield with extended use, and the decrease in activity for the O_R-C₃N₄ as compared to the O_T-C₃N₄ and unmodified samples, oxidation of g-C₃N₄ was considered as an alternative source of CO observed.

Therefore, samples were analyzed before and after use by XPS and FTIR to determine the degree of oxidation consistent with this hypothesis.

The structural and functional groups reported for g-C₃N₄ are given in Table 1. FTIR analysis was used to determine if any changes to the functional groups following use as a photocatalyst (Figure 9). All samples showed the finger printing region characteristic for heptazine subunit and functional groups consistent with expected positions for g-C₃N₄ as detailed in Table 1.

Table 1. Peak positions of structural and functional groups observed by FTIR spectra for g-C₃N₄ and O-C₃N₄ samples.

FTIR analysis		Reference
Wavenumber, cm ⁻¹	Functional group	
1250-1324-1405-1454-1571-1636	C ₆ N ₇ heptazine heterocycle ring	32, 45-52
807-813	triazine ring stretching	32, 45-52
892	Cross link heptazine deformation	45
1654	N-H deformation	46, 50
1610	C=N	47, 51, 52
3469-3420	NH ₂ stretching.	46, 47, 50, 52
3337-3137	asymmetric and symmetric stretching N-H	45-47, 51, 52
3500-3100, 3350	OH	32, 58, 59

A decrease in the absorbance associated with the g-C₃N₄ functional groups was observed following use in the photoreactor (see Figure 9). Such changes were also analyzed by XPS.

Figure 9. Diffuse FTIR analyses for each sample used and unused.

The XPS elemental profiles and expected position of peaks related to structural and functional groups have previously been reported by researchers investigating g-C₃N₄, as detailed in Table 2. Using this information, in conjunction with analysis of samples before and after photocatalytic treatment, changes to the photocatalyst surface during CO₂ reduction are reported.

Table 2. Structural and functional group reported peak positions observed in XPS for g-C₃N₄ and O-C₃N₄

XPS Analysis			
Element	Kind of interaction	Binding energy, eV	Reference
N	400		32, 37, 45-52, 58, 60
	N1(N sp ²) C-N=C	398.5	32, 37, 45, 47-51, 58
	N2 (N-H)	399.5	37, 45, 51
	N3 (sp ³) NC ₃	400.2	32, 37, 45, 47, 49, 50, 58
	N4 (NH ₂)	401	32, 45, 48, 49, 58
	<i>N-O, or due to O₂ and H₂O adsorbed</i>	<i>401.8</i>	51
	Nπ-π	404.2	37, 45, 50
C	284.8		32, 37, 45-52, 58, 60
	C-C	284.6	32, 37, 45-47, 49, 58

	C-CH ₂ , C-OH, C-O	286.2	37, 45, 46, 48, 58
	C-N	287.2 (286.2 ref 8)	32, 45
	C1	288.2	37, 45, 47-49, 58
	C2	288.8	45
	N-C=O	289.8 (288.9 ref 8)	32, 45, 46, 58
	C π - π	293.8	45
O	532.6		32, 37, 45-48, 58, 60
	C=O	530	61
	C-OH, O-C-N	531.9	32, 36, 46
	H ₂ O adsorbed	532.9	37, 46, 48
	O-N	533.4	32

Theoretically, XPS should show a C/N molar ratio of 0.75 for g-C₃N₄. Investigation of functional groups, particularly carbon oxygen functionalities, that could alter this ratio are detailed in Table 2. To understand the XPS signal, one must consider the differing bonding configurations that both nitrogen and carbon adopt in the g-C₃N₄ structure (Figure 10). Essentially, carbon has two unique bonding configurations (C1, C2) whilst nitrogen adopts four unique configurations (from N1 to N4) in the carbon nitride structure.⁴⁵ This causes a splitting in the XPS signal for each configuration allowing investigation of structure.

Figure 10. Structure of the repeating heptazine carbon nitride unit with the corresponding atomic assignment according to XPS analysis.

Both elements C and N, in the C_3N_4 graphitic structure, generate the signal due to the π to π^* transition (shake-up) at around 293.8 eV and 404.2 eV respectively, as reported in Table 2. In addition to chemical shifts in XPS due to different bonding configurations of carbon and nitrogen in perfect C_3N_4 crystals, the peak positions due to oxygen functionalities characteristic of $O-C_3N_4$ samples are also reported in italic characters in Table 2.

Analysis of elemental composition via XPS was initiated with a wide energy survey (see Figure S3), performed in three locations, to look for contamination and estimate differences in the elemental constitution. Only carbon, nitrogen and oxygen peaks of significant magnitude for quantification were observed. The value of binding energy for each element observed showed good agreement with literature (see Table 2).

This analysis showed that the quantity of oxygen present in the pure as prepared C_3N_4 is the lowest of all the samples. However, for samples that had undergone thermal annealing and/or testing as a photocatalyst, the oxygen content was increased in comparison to untreated material.

Table 3. Survey analysis average of three (wt%) XPS measurements of used $g-C_3N_4$, unused $g-C_3N_4$ and $O_R-C_3N_4$ reference.

	$O_R-C_3N_4$	$g-C_3N_4$	$g-C_3N_4$ used	$O_T-C_3N_4$	$O_T-C_3N_4$ used
O	5.4	2.0	3.0	2.9	3.6
C	49.6	48.6	49.9	48.5	48.0
N	44.9	49.4	46.8	48.6	48.4

By comparing changes in carbon and nitrogen percentages across sample groups, oxygen inclusion during synthesis of $O_R-C_3N_4$, thermal treated $O_T-C_3N_4$ or used $g-C_3N_4$ data suggests that oxygen incorporated in the structure is likely substituting in one or more of the differing nitrogen positions N1, N2 and N3. In fact, a loss in nitrogen concurrent with increases in oxygen percentage can be observed in Table 3. To discriminate between contributions from strongly adsorbed contaminants and

different functional groups, high resolution scans of N, C and O regions were measured (see figure S4) and deconvoluted based on assignments reported in Table 2.

For all samples analyzed, used and unused, a large proportion of the oxygen signal observed came from the water strongly bound to the catalyst. In the as prepared g-C₃N₄ sample, all the oxygen observed was from adsorbed contaminants, H₂O and CO₂. Analysis of the g-C₃N₄ sample, thermally and chemically doped samples showed increased contributions from oxygen in lattice (N-C-O-C) and oxygen functionalization of periphery aromatic regions (ArC=O), as detailed in Table 4.

Table 4. Contributions from different oxygen carbon and nitrogen functionalites (Wt %) estimated from deconvolution of XPS each peak centred about at 288.4, 531.5 and 398.9 eV, respectively.

		Wt %				
		O _R -C ₃ N ₄	g-C ₃ N ₄	g-C ₃ N ₄ used	O _T -C ₃ N ₄	O _T -C ₃ N ₄ used
Binding energy, eV	O Type	Oxygen				
529.92	O=C Ar	1.37	0.00	0.00	1.35	2.38
531.15	O-C-N in lattice	8.63	0.03	2.34	1.57	2.59
532.28	OH water, CO ₂	90.00	99.97	97.66	97.18	95.03
Binding energy, eV	N Type	Nitrogen				
398.72	N1 edge	68.23	71.34	64.47	65.28	66.88
398.83	N2 external	8.43	3.77	6.83	10.29	7.50
400.08	N3	8.47	9.10	11.12	8.57	10.76
401.08	N4 external	12.79	12.97	14.58	12.90	11.92
404.25	π-π	2.09	2.82	3.01	2.96	2.97
Binding energy, eV	C Type	Carbon				
284.82	C-C	27.49	22.57	27.58	25.29	25.75
286.4	C-OH	5.43	3.30	3.74	1.81	1.83
288	C1	40.82	40.75	29.11	46.63	35.66
288.6	C2	22.28	28.76	33.81	21.33	30.92
289.64	C=O external	1.27	1.74	2.06	1.00	1.94
293.37	π-π	2.72	3.37	3.71	3.96	3.91

This is suggestive that UV-Vis irradiation of g-C₃N₄ in the presence of water leads to photocorrosion and oxidation of the g-C₃N₄. To understand the changes in the carbon nitride structure, due to increasing oxygenation, the nitrogen and carbon regions were also investigated. In the nitrogen region, due to the bonding in the C₆N₁₀H₁₂ heptazine monomer, nitrogen adopts four bonding configurations labelled N1 to N4 in Figure 10, each with a unique binding energy detailed as reported in literature in Table 2 and experimental data found in Table 4. N1 are present on the edges of heptazine base units, N3 are present at the center of a perfect heptazine base, whilst N2 type link heptazine sub units together and N4 represent the groups left upon the termination of polymerization. In a heptazine monomer (Figure 11a) the ratio of N1:N3:N4 is 6:1:3, N2 does not appear in the base unit melem; however as the monomer polymerizes, some N4 type are converted into N2 with the ratio of N4:N2 groups dependent on the geometry of polymerization (Figure 11c). In the g-h heptazine phase under study here, confirmed by XRD and UV-Vis (see above), heptazine monomers polymerize along parallel zig zag chains linked by single N4 units as illustrated in Figure 11e. Polymerizing in the g-h heptazine structure results in changes in the ratio of N2 to N4 nitrogen, relative to the monomer, as one N4 is replaced with one N2 for every unit extension along the polymerization axis. Illustration of the changes in nitrogen group bonding during polymerization is detailed in Figure 11.

Figure 11. Pathway from Melem unit (a) and dimer (b) to polymeric $g\text{-C}_3\text{N}_4$ in two different geometry base unit, sheet (c) and firstly linear chain (d) and then in zig zag conformation (e).

Table 5. Change in nitrogen ratios as a function of chain length for h-g-heptazine structure.

	N1	N2	N3	N4	N1 %	N2 %	N3 %	N4 %
Dimer	12	1	2	4	63.16	5.26	10.52	21.05
Tetramer	12	1	2	3	66.66	5.55	11.11	16.67
Octamer	12	1	2	2.5	68.85	5.71	11.43	14.28
Infinite chain	12	1	2	2	70.58	5.88	11.76	11.76

As reported in Table 5, the short heptazine chains result in $N1 > N4 > N3 > N2$ ratio, whereas at the infinite chain lengths ratios become $N1 > N4 = N3 > N2$ which is equivalent to a $N1:N2:N3:N4$ ratio of 12:1:2:2. This corresponds to 70.58:5.88:11.76:11.76 percentage ratio useful when comparing this perfect crystal to the percentages obtained from XPS.

For the synthesized g- C_3N_4 , the $N1:N2:N3:N4$ percentage ratio from XPS was 71.34:3.77:9.10:12.97, which is good match for expected N1 % in g-h-heptazine with largely perfect heptazine base units. Considering N4 suggests long range but not infinite chains consistent with XRD, the excess of N4 percentage may also be due to unreacted precursor / monomer material or some shorter chains. Deviation, a lower % than expected, in N2 suggests crystal imperfections, defects or branching of the chains removing side N2 groups, whilst contemporaneously increasing N4 groups. Likewise, the N3 ratio was lower than expected for perfect crystals, suggesting vacancies in some of the heptazine blocks.

$N1:N2:N3:N4$ ratio from carbon nitride oxidized during synthesis, thermally or during photocatalysis are compared in Table 4. All treatments were observed to result in significant reduction in N1 side groups, relative to the starting material, suggestive of damage/distortion to the base of the heptazine unit. Deconvolution of the N2 and N4 (edge and terminal groups) in used g- C_3N_4 shows an increasing N4 percentages, suggesting a reduction in average chain length. In the $O_T-C_3N_4$ sample, higher N2 and N4, with lower N1 ratios than expected for a perfect g-h-heptazine structure were observed, presumably due to oxygen disrupting periodicity. Its use as a potential photocatalyst showed further changes in the $N1:N2:N3:N4$, ratio which is consistent with additional restructuring of the polymer towards a more open arrangement.

Deconvolution of XPS C1s region was also used to look at changes in structure following photocatalysis. In this region, the C-C and C-OH peaks are characteristic of adsorbed species or deviance from heptazine structuring / reformation. Both thermal oxidation and use as a potential photocatalyst increased the C-C peak indicative of a material with less heptazine/g- C_3N_4 present.

Likewise, signals due to C=O bonding were observed to increase in both sample sets following use as photocatalysts, which is consistent with increasing counts in the oxygen region observed for the same treatment. C1 and C2 regions represent differing carbon nitrogen bonding. In all cases, it showed decreasing C1 and increasing C2 content following photocatalytic use due to changes in nitrogen content during use.

Intentional oxygen doping and repeated use in CO₂ photocatalytic reduction experiments both led to reduced yields of CO in subsequent tests and simultaneous increases in the oxygen content of the g-C₃N₄. Also, UV-Vis irradiation was observed to give rise to changes in the g-C₃N₄ structure (N1:N2:N3:N4 ratios).

Another factor to be considered is the role of N in the structure. In particular, as it can be noticed in Table 3 (XPS survey analysis), the increase of O content from the unused to used g-C₃N₄ sample and to O doped samples (O_T-C₃N₄ to O_R-C₃N₄) corresponds to a decrease of the total content of N. This % variation which could be due to a replacement of N with O in the structure, could alter CO₂ adsorption properties adversely.

In photocatalytic tests carried out in the presence of CO₂ under the experimental conditions used (pressure, temperature, and light intensity), the increase in the O content of the g-C₃N₄ was caused by photocorrosion. It is not easy to determine if H₂O, CO₂ or both are the source(s) of O observed in this work. The photocorrosion means the modification of the C₃N₄ structure, due to the substitution of N with O during the photoreaction, as indicated by XPS measurements (see Table 3) and reported also by Huang et al when H₂O₂ was used to dope C₃N₄ with oxygen.⁴⁶ The mechanism which led to the final modified C₃N₄ in our experimental conditions could give rise to formation of a non-stoichiometric or defective solid. Moreover, the figures related to the percentage of the various N types and shown in Table 3 strongly suggest a rearrangement of the g-C₃N₄ structure following use in photocatalytic testing.

Figure 12. Schematic illustration of N replacement by O in g-C₃N₄ material, where red and black dots indicate N and C element, respectively, replacing with O.

As far as the runs carried out in the absence of CO₂ are concerned, the presence of CO₂ adsorbed on the catalyst surface or as impurity in the system cannot be excluded and it can be compatible with the CO formation.⁴⁴

In current literature, only a few papers report repeat testing and generally they concern runs carried out using C₃N₄ composites and or bulk materials with low surface area. Qin et al,⁴⁰ observed that in the presence of barbituric acid-g-C₃N₄ composite the photoreduction of CO₂ to CO and H₂ was quite reproducible, at least after three catalytic consecutive tests. Probably, in this case, the presence of barbituric acid stabilized the g-C₃N₄ structure avoiding the replacement of N by O and consequently no decreasing activity was observed. A similar stabilizing effect due to the presence of ceria can be hypothesized in the case of CeO₂-C₃N₄ composite that showed a constant CH₄ and CO production after 4 runs as reported from Li et al.²⁹

In contrast with the present study, Fu et al. reported reproducible photoreduction of CO₂ with both bulk g-C₃N₄ and O-C₃N₄ photocatalysts. However, it is important to note that the amount of CO₂ photoreduction products observed by Fu et al. was always lower than the lowest one revealed in our study. Moreover, no structural studies were carried out on the g-C₃N₄ photocatalyst following use for

CO₂ reduction.³² Interestingly, in that work no CO formation was observed during control experiments in the absence of CO₂.

4. Conclusions

During testing of g-C₃N₄ and oxygen doped g-C₃N₄ as the photocatalysts for CO₂ reduction, the main product detected was CO. No other products were detected. The pure material was observed to be more active than the oxygen modified analogue for CO production. A reduction in the rate of CO generation, under irradiation, was observed during both single runs and repeated test cycles with the same catalyst. During control tests in the absence of CO₂, the production of CO was observed under irradiation, raising questions about the origin of the CO and the stability of g-C₃N₄.

Diffuse reflectance FT-IR analysis showed the expected functional groups characteristic of g-C₃N₄ present in all samples. Functional group peak intensity declined with oxygen doping or photocatalytic use. Given the qualitative nature of FTIR, samples were further investigated with XPS to quantify the changes in the functional groups during photocatalytic use or intentional oxidation.

FTIR and XPS analysis showed that the thermal and the photocatalytic treatments produce similar changes to the functional groups, consistent with oxidation of the C₃N₄ structure. The oxidation of the g-C₃N₄ under UV-Vis irradiation was slightly different to the one obtained with oxygen doping, but produced the same effect on catalytic behavior, i.e. a reduction in product yield under irradiation.

The XPS data showed an increase in oxygen content, both absolute and bound to carbon, after thermal treatment and after photocatalytic use. This study allowed the examination of where and how oxidation disrupts the material structure, showing changes to the heptazine base units along with changes in bonding between units following oxidation or photocatalytic use.

This work points that under our reactor conditions, the photoexcitation actually leads to oxidation of C_3N_4 rather than water splitting, leading to a loss of photocatalytic activity i.e. photocorrosion. Given that the valence band edge of g- C_3N_4 is close to the water oxidation potential, this material has limited over-potential to drive water oxidation and as such, may accumulate intermediate ROS species that may oxidize the catalyst, changing its activity as observed here. C_3N_4 may still be useful as a photocatalyst; however, the use of a z-scheme or a second co-catalyst material on which to run the oxidation reaction may be necessary to avoid g- C_3N_4 oxidation. Many researchers use a z-scheme consisting of g- C_3N_4 in composite materials such as g- C_3N_4 - TiO_2 where the latter material performs the water oxidations steps, decreasing the probability of g- C_3N_4 oxidation.

Supporting Information

Measurement of irradiance from 100W Xe lamp with IR filter, representative XPS survey, C, N and O XPS analysis and the quantum yield of all samples are reported. This material is available free of charge via the Internet at ...

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